



The Sensitivity of V₂O₅:Sm₂O₃ Thin Films against NO₂ Toxic Gas

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Abstract

Vanadium oxide V₂O₅ thin films with variation doping ratios of Sm₂O₃ (2, 4, 6, and 8 % wt.) on corn glass and p- type silicon substrates were prepared by pulsed laser method. The X-ray diffraction peaks for V₂O₅ decreases with doping ratio of Sm₂O₃. FESEM images for V₂O₅ and doped thin films illustrates clusters with a homogeneous distribution in nano scale. The energy gap varied upon the increment of doping concentration, starting from 2.610 eV to 2.7 eV. Gas sensor measurement of pure and doped V₂O₅ demonstrated a sensitivity to NO₂ gas, and the sensitivity expanded upon the increment of operation temperature. The greatest sensitivity was found to be about 99%, while best response time of 10s and recovery time of 18s were recorded using the 4% Sm₂O₃ sample at 50 °C.

Key Words: Thin Film, Nano, Structures, Gas Sensor.

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Introduction

Many oxidation states of vanadium give rise to a wide variety of oxides. The oxides of the claimed vanadium have a complex stoichiometric composition. Concerning this research, vanadium pentoxide V₂O₅ has a large application among those metal oxide semiconductors, vanadium pentoxide draws huge enthusiasm through as far back as decades owing should its broad assortment of utilizations. Its multivalency, layered structure, optical band gap, chemical and thermal stability, very good thermoelectric characteristic, and so on are the properties that make activate vanadium pentoxide V₂O₅ an ensuring material for demands in microelectronics, electrochemical and optoelectronic appliance. It could a chance to be utilized concerning illustration a gas sensor, a window for solar cell and electrochromic gadgets and in addition optical switches, yet all there need been a developing enthusiasm toward the creation of thin film batteries (TFB) (Benmoussa et al., 2002). V₂O₅ is the greater part stable oxide furthermore

indicates semiconductor characteristic for an energy gap of 2.2 eV (Zou and Gao 2010). There are two bulk structures for vanadium pentoxide, the first one is α-V₂O₅ and the second one is γ-V₂O₅. The structure of α-V₂O₅ is orthorhombic layered structure. Pyramid structural plane expands for five oxygen molecules encompassing one V molecule (Londero and Schröder 2010; Ali et al., 2018). Samarium oxide (Sm₂O₃) nanoparticles, prominently known as SM-MITE powders, are accessible in covered and hydrophilic forms. Samarium belongs to block F, Period 6 while oxygen belongs to block P, Period 2 of the periodic table. Gas sensor examination of V₂O₅:Sm₂O₃ thin film with concentration (2,4,6,8) wt.% were prepared using pulsed laser deposition on glass. This examination depends on the adsorption of gas molecules on the surface of the material which increase with increasing of surface to volume ratio (Bhushan 2007).

69

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The sensitivity is determined by the change in resistance when the sensor is exposed to a certain gas concentration. The sensitivity can be determined from (Bochenkov and Sergeev 2010):

$$S = \left| \frac{R_g - R_a}{NR_a} \right| \times 100\% \quad (1)$$

where (R_a) and (R_g) are electric resistance of sensor over air and in presence of gas respectively, (N) will be the gas concentration. The response time of the sensor is characterized by change that takes those sensors should to achieve 90% of max/min worth of resistance upon introduction of the reducing /oxidizing gas (Gibbs et al., 2013).

Experimental

V₂O₅ nanoparticles with different doping concentrations of Sm₂O₃ (2, 4, 6 and 8) wt.% with high purity of 99.999 % is pressed under pressure of 5 tons to form a target of 1.5 cm diameter and 0.2 cm thickness. It should be as dense and homogeneous as much as possible to ensure a good quality of the deposit. In the Pulsed Laser Deposition (PLD) technique, which is a setup of 200 mJ with constant shoot of 600 are carried out. The distance between the target and the substrate is 2 cm, while the distance between the target and the laser source is 12 cm, the laser may be kept tabs onto a pivoting target the place it evaporates those material to structure a plume. The plume goes Previously, possibly vacuum or gas foundation area the middle of those target and the substrate When depositing on the substrate to structure a film. Those methodology is preceded toward those clearing of the chamber to a high vacuum, those PLD method need been demonstrated on a chance to be a successful system should store high-quality films, Uniform thin films, The intricate material films what's more secondary controllability of arrangement of dainty thin films, (Isam et al., 2018). V₂O₅:Sm₂O₃ thin film have been readied by PLD system. The pulsed laser deposition with a vacuum chamber generally at (10⁻³ Torr) vacuum conditions, the target is putted Infront to the Nd: YAG Q - switching laser beam with angle 45°. The thickness about of 100 nm where measurements by weighting method. The samples then annealed up to 400°C for one hour in the vacuum oven.

The structural and morphological experiments are containing X-ray diffraction, field emission scanning electron microscope X-ray diffractions using a Philips X-ray diffractometer system. The X-ray diffraction (XRD) pattern of the (V₂O₅) and

(V₂O₅:Sm₂O₃) film deposited on Si substrate toward SHIMADZU XRD -6000 X-ray diffract meter [CuK_α radiation λ=0.154 nm] done 2θ range from on. The interplaner distanced d_{hkl} for different planes are measured.

Field Emission Scanning Electron Microscopy (FESEM) Vega TOSHIBA S 4160 where used for measuring the prepared samples. The absorption spectrum for V₂O₅ with different doping centralization of Sm₂O₃ films once glass substrate is measured utilizing OPTIMA SP_3000 UV_VIS spectrophotometer covering a range from 190 to 1100 nm utilizing glass substrates as a reference gas sensor detector namely, reaction time also recuperation run through. To this end as setup demonstrated.

Those varieties of the sensor resistance when presented to air_NO₂ gas ratio volumetric concentration of 10% gas to air ratio.

The test gas through a flow meter, furthermore needle valve plan. The blending gas is fed may be fed through a tube again those sensors inside those test chamber to give acceptable those true sensitivities.

Results and Discussion

Figure 1 indicates the X-ray diffraction patterns of deposit V₂O₅ thin films on Si substrates with orientation (111) by PLD technique with different doping concentrations of Sm₂O₃ (2, 4, 6 and 8) wt.%, we have to mention here that we used the Si substrate to get more crystallinity than amorphous substrate. Hall measurements show that all these films have a negative Hall coefficient (n type charge carriers), this is attributed may be due to the number of electrons excited above the conduction band mobility edge is larger than the number of holes excited below the valance band mobility edge. The peak of Si located at 2θ= 28.1 ° which related (111) direction. At (2 and 4) wt. % Sm₂O₃ tests their two peaks spotted at (2θ= 32° with 41°) which correspond to (1̄11),(301) direction, respectively for Diffraction information ICDD (96-101-0341), for V₂O₅ crystals, same time there would different peaks show up In (6, and 8) wt. % Sm₂O₃ at (2θ=45.0259°, and 51.2468°) alluded will {(1̄03) and (020)} direction, respectively, to Sm₂O₃ crystals.



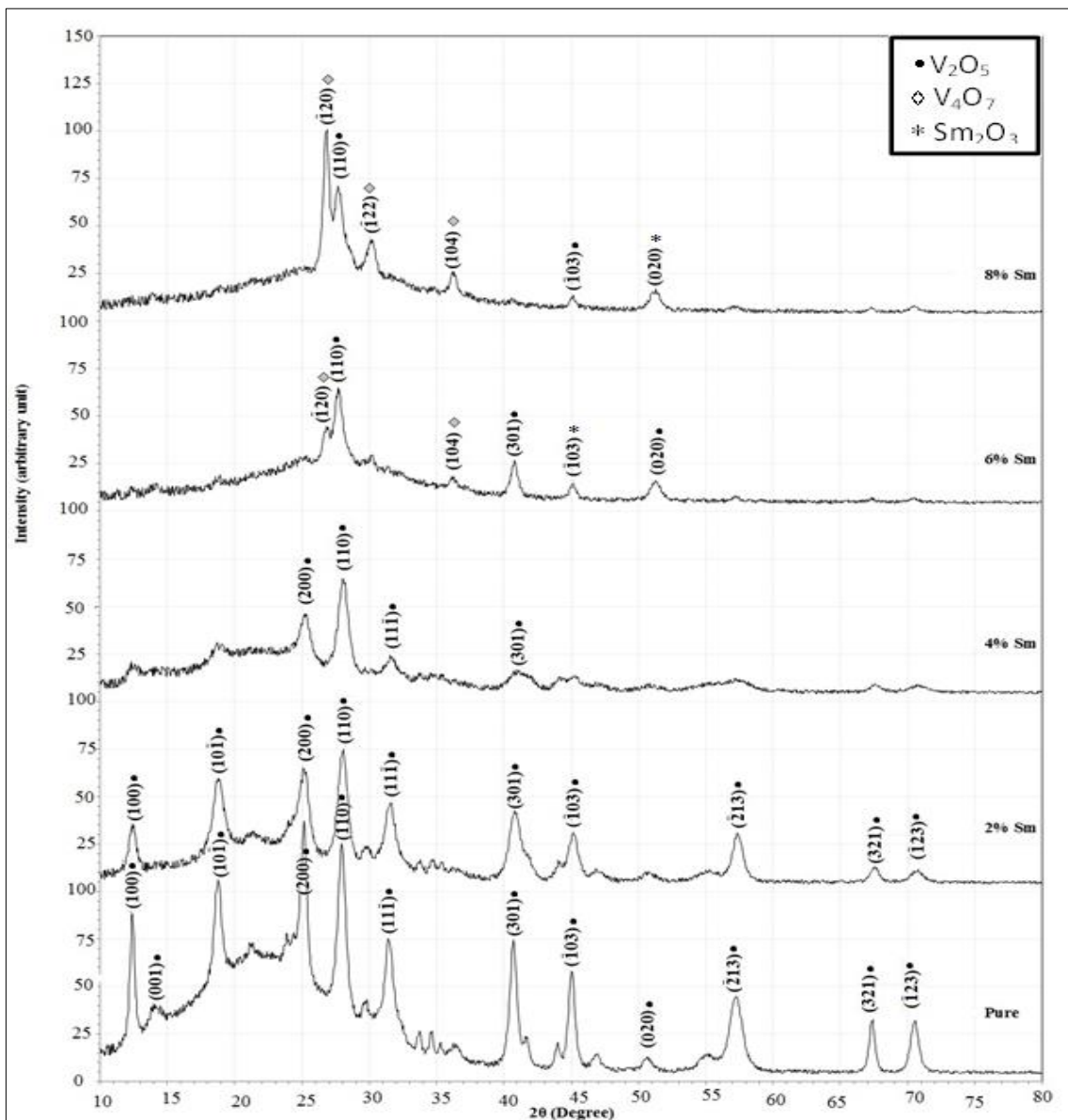


Figure 1. XRD for V₂O₅: Sm₂O₃ at annealing temperature of 400 °C.

The x-ray information of thin films distinguishable with that for known V₂O₅ monoclinic structure and Sm₂O₃ cubic structure as stated by universal centers for Diffraction information ICDD (96-210-2584). this expanding of different doping concentrations of Sm₂O₃ lead to decrease the V₂O₅ peak intensity, as well as, the Sm₂O₃ peaks would be showing up during high concentrations. Table (1) Shows the crystalline size and hkl of Sm₂O₃ doped V₂O₅. At the doping to 6%, two peaks with a low intensity at about (2θ) equivalent to: 26.795° and 36.250° for the planes (120) and (104) with property of V₄O₇, as indicated by the ICDD(NO.96-100-8026), which denote that influence of doping has prompted the

development of new phase structure in V₄O₇ films and the atoms of doped material have effectively come in the Vanadium Oxide host lattice. (Zhao et al., 2011).



Table 1. Values of X-ray diffraction peaks for V₂O₅ films with distinctive doping concentrations of Sm₂O₃ at 400 °C.

Sm ₂ O ₃ %	2θ {Deg.}	FWHM {Deg.}	d _{h.k.l} Exp.{Å}	C-S {nm}	h.k.l	D _{h.k.l} Std.(Å)	phase	Caed no.
Pure	12.389	0.416	7.138	19.2	(100)	7.114	V ₂ O ₅	96-210-2584
	14.059	1.111	6.294	7.2	(001)	6.284	V ₂ O ₅	96-210-2584
	18.747	0.626	4.729	12.9	(10-1)	4.712	V ₂ O ₅	96-210-2584
	25.158	0.485	3.536	16.8	(200)	3.557	V ₂ O ₅	96-210-2584
	27.962	0.690	3.188	11.9	(110)	3.192	V ₂ O ₅	96-210-2584
	31.469	0.691	2.840	11.9	(11-1)	2.846	V ₂ O ₅	96-210-2584
	40.710	0.625	2.214	13.6	(301)	2.217	V ₂ O ₅	96-210-2584
	45.039	0.553	2.011	15.5	(-103)	2.008	V ₂ O ₅	96-210-2584
	50.651	0.761	1.800	11.5	(020)	1.611	V ₂ O ₅	96-210-2584
	57.240	1.110	1.608	8.2	(-213)	1.611	V ₂ O ₅	96-210-2584
67.354	0.485	1.389	19.7	(321)	1.391	V ₂ O ₅	96-210-2584	
70.518	0.693	1.334	14.0	(-123)	1.335	V ₂ O ₅	96-210-2584	
2%	12.420	0.631	7.120	12.7	(100)	7.114	V ₂ O ₅	96-210-2584
	18.794	0.949	4.717	8.5	(10-1)	4.712	V ₂ O ₅	96-210-2584
	25.221	0.735	3.528	11.1	(200)	3.557	V ₂ O ₅	96-210-2584
	28.033	0.857	3.180	9.6	(110)	3.192	V ₂ O ₅	96-210-2584
	31.548	0.822	2.833	10.0	(11-1)	2.846	V ₂ O ₅	96-210-2584
	40.813	0.947	2.209	8.9	(301)	2.217	V ₂ O ₅	96-210-2584
	45.152	0.839	2.006	10.3	(-103)	2.008	V ₂ O ₅	96-210-2584
	57.383	0.872	1.604	10.4	(-213)	1.611	V ₂ O ₅	96-210-2584
	67.523	0.735	1.386	13.0	(321)	1.391	V ₂ O ₅	96-210-2584
	70.695	1.051	1.331	9.3	(-123)	1.335	V ₂ O ₅	96-210-2584
4%	25.256	0.843	3.523	9.7	(200)	3.557	V ₂ O ₅	96-210-2584
	28.072	0.896	3.176	9.1	(110)	3.192	V ₂ O ₅	96-210-2584
	31.592	0.860	2.829	9.6	(11-1)	2.846	V ₂ O ₅	96-210-2584
	40.870	1.281	2.206	6.6	(301)	2.217	V ₂ O ₅	96-210-2584
6%	26.795	0.598	3.324	13.7	(-120)	3.327	V ₄ O ₇	96-100-8026
	27.680	0.643	3.220	12.7	(110)	3.192	V ₂ O ₅	96-210-2584
	36.250	0.631	2.476	13.2	(104)	2.478	V ₄ O ₇	96-100-8026
	40.699	0.518	2.215	16.3	(301)	2.217	V ₂ O ₅	96-210-2584
	45.025	0.459	2.011	18.7	(-103)	2.008	V ₂ O ₅	96-210-2584
51.246	0.920	1.781	9.6	(020)	1.611	V ₂ O ₅	96-210-2584	
8%	26.795	0.500	3.324	16.3	(-120)	3.327	V ₄ O ₇	96-100-8026
	27.680	0.675	3.220	12.1	(110)	3.192	V ₂ O ₅	96-210-2584
	30.167	0.683	2.960	12.0	(1-22)	2.970	V ₄ O ₇	96-100-8026
	36.250	0.631	2.476	13.2	(104)	2.478	V ₄ O ₇	96-100-8026
	45.180	0.560	2.005	15.4	(-103)	2.008	V ₂ O ₅	96-210-2584
	51.246	0.920	1.781	9.6	(020)	1.611	V ₂ O ₅	96-210-2584

Figure. (2) indicates FESEM images to V₂O₅ thin films at different Sm₂O₃ concentrations 0%,2%,4%,6%, and 8% wt.%. Every last one of images seemed in fig. (2) appears homogeneous group distribution with columnar structure while for pure V₂O₅ image the structure revealed some holes; where the average grain size values decrease slightly with increasing doping. The significant change in the average value of cluster surface with

increasing Sm₂O₃ proportion. The least cluster surface showed up maximum doping proportion 8% while maximum cluster surface appeared to least doping proportion 2%. The no. of clusters per image appears reversible conduct comparing to the average of cluster surface with increasing of doping. These behaviors indicate increasing the surface of cluster boundaries where the grains didn't bring enough energy to enucleation to decrease cluster



area surface. Expanding Sm₂O₃ proportion prompts increment number of Sm₂O₃ molecules on the substrate surface which act as traps for add atoms causing to decrease grain size by extent toward expanding amount about nucleation centers

prompts making another phase from vanadium oxide likewise. It is clear from the figure (2) that the particle size decreases with the increase of samarium oxide concentration.

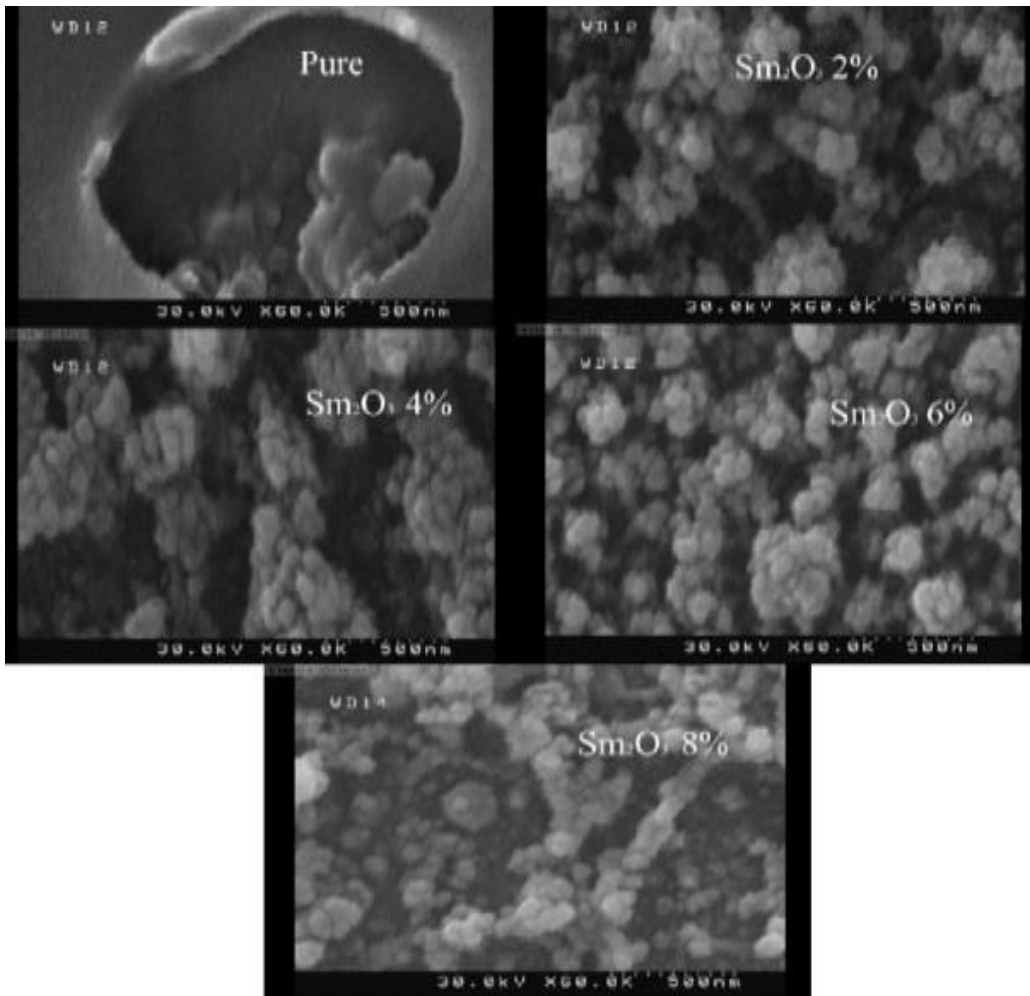


Figure 2. FESEM for V₂O₅:Sm₂O₃ with different Sm₂O₃ ratios

The optical band gap (E_g^{opt}), which might assistance evaluate the electronic what's more thermoelectric characteristic about materials (Isam et al., 2018). The optical energy gap (E_g^{opt}) have been evaluated utilizing Tauc equation by the extrapolation of the parcel at $(\alpha h\nu)^2 = 0$. The energy gap expanding with expand of doping concentration starting with 2.610 eV to 2.7 eV as shown in Figures (3). It observed that (E_g^{opt}) will be expanding marginally, furthermore shifting towards the ultra-violate- range, this may be due to the impact from impurity or disorder. Also, whatever available defects, semiconductors prompt

to local electric fields that influence the band tails close to the band edge. Likewise, the measure impact to nanostructure produce this shift. The results of energy gap show that the prepared thin films have localized states which result from the density of defects at the grain boundaries furthermore donor levels, along these lines we might that those optical energy gap could a chance to be controlled through impurities proportions and the particle size of nanostructure, in addition the variation of the crystallite size expanding of energy gap place by the effect of quantum confinement.

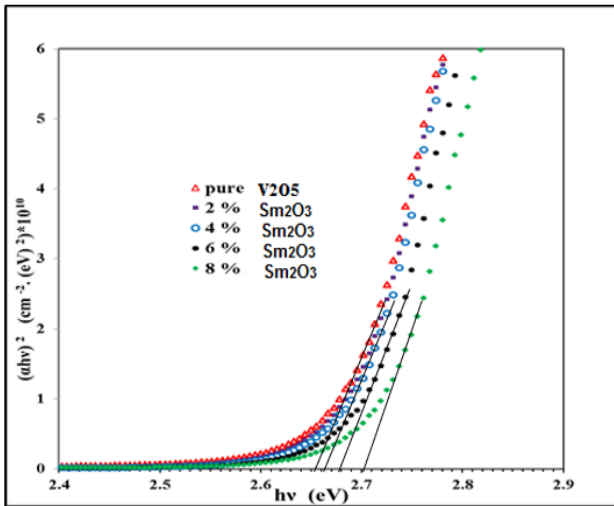


Figure 3. $(\alpha hv)^2$ against photon energy (hv) for V₂O₅ films at (2, 4, 6, and 8) wt. % Sm₂O₃ proportion

Figure (4,5,6 and 7) indicates the sensing conduct about pure V₂O₅ and doping with Sm₂O₃ /Si for NO₂ gas, it is clear from this figure that the resistance increases with gas on till reach the maximum value and the behavior will be opposite for gas shutting. From figures it is found that greatest sensitivity about 99%, and the response time equivalent to 10s, while, recovery time 18s for 4% Sm₂O₃ sample at 50 °C In general, the mechanism of V₂O₅ is identified with the ionsorption for gas species through the surface, prompting to charge exchange of the gas and surface. Particles Moreover progressions in the electrical conductance (Filipovic et al., 2013).

Figures show that the increases in the resistance value when their films presentation to NO₂ gas, {Gas ON}, After that the resistance again sloping at the shutdown of the gas {Gas OFF}. This behavior can be attributed to the subsequent (NO₂) gas that undergoes an ionic reaction with surface adsorption oxygen, these electrons on oxygen can be concentrated starting from the semiconductor materials, the electrical conductivity of V₂O₅ materials will decrease causing the increasing in resistance (Filipovic et al., 2013). The variable resistance value indicated by this metal oxide sensor in the air, oxygen is adsorbed on the surface, besides the formation of dissociates (O⁻), the place where the electron is on the oxygen can be concentrated starting from these semiconductors. This electron resorts to increasing the resistance of n-type semiconductors, where electrons are the main carriers. For gas (NO₂), nitrogen reacts to adsorbed (O⁻) to form NO₂, and an electron can be reintroduced under these semiconductors, resorting to decreasing resistance. The adversary acts on the middle of the oxygen, removing electrons, and the combustible gas restores these electrons. Thus, the consistent quality of resistance of the metal oxide depends on the fixation of the flammable gas. However, it can be seen from the table below that the addition of samarium to vanadium oxide at any concentration does not have a significant effect.

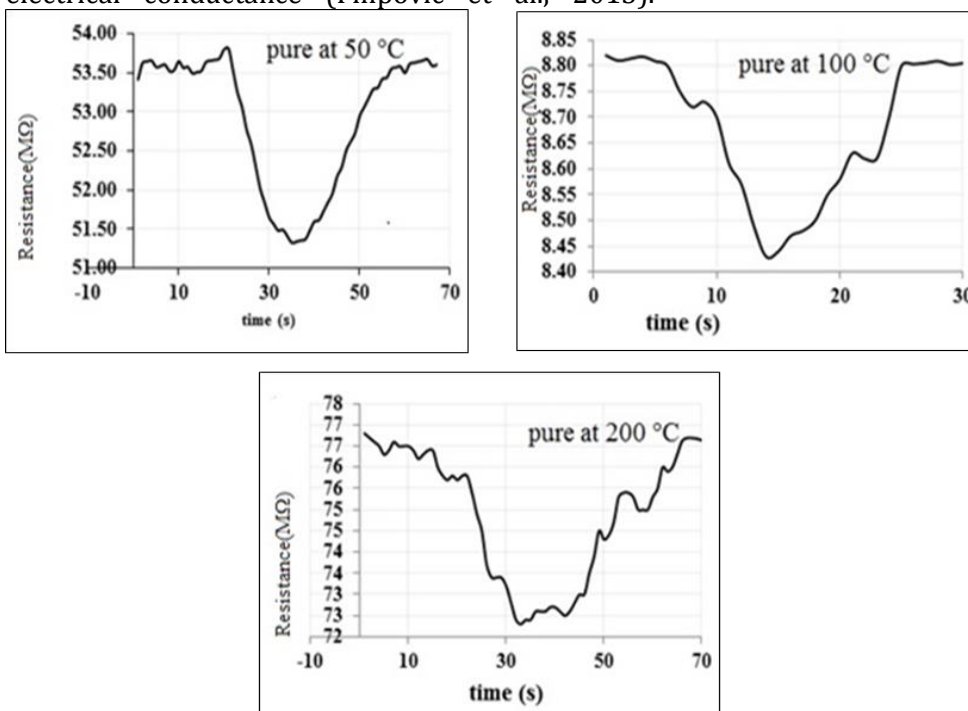


Figure 4. Variety for resistance as a function to the time pure V₂O₅ films at 50,100 and 200 °C



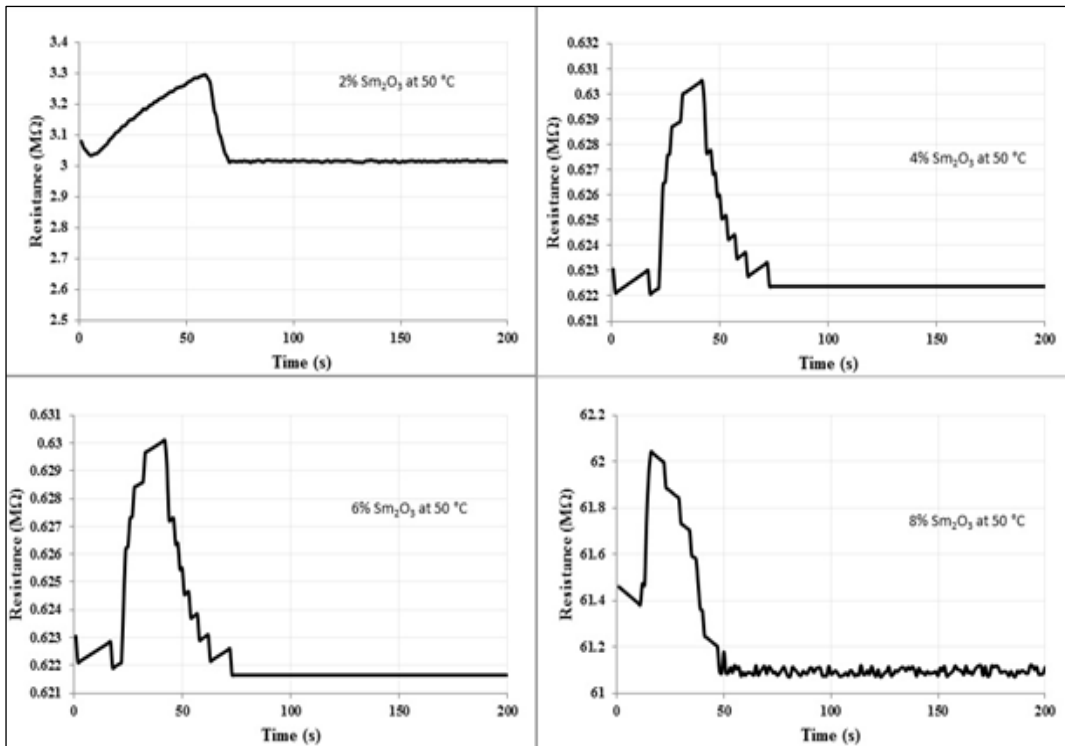


Figure 5. Variety for resistance as a function of time V₂O₅ films with (2, 4, 6, and 8) wt. % Sm₂O₃ ratio at 50 °C

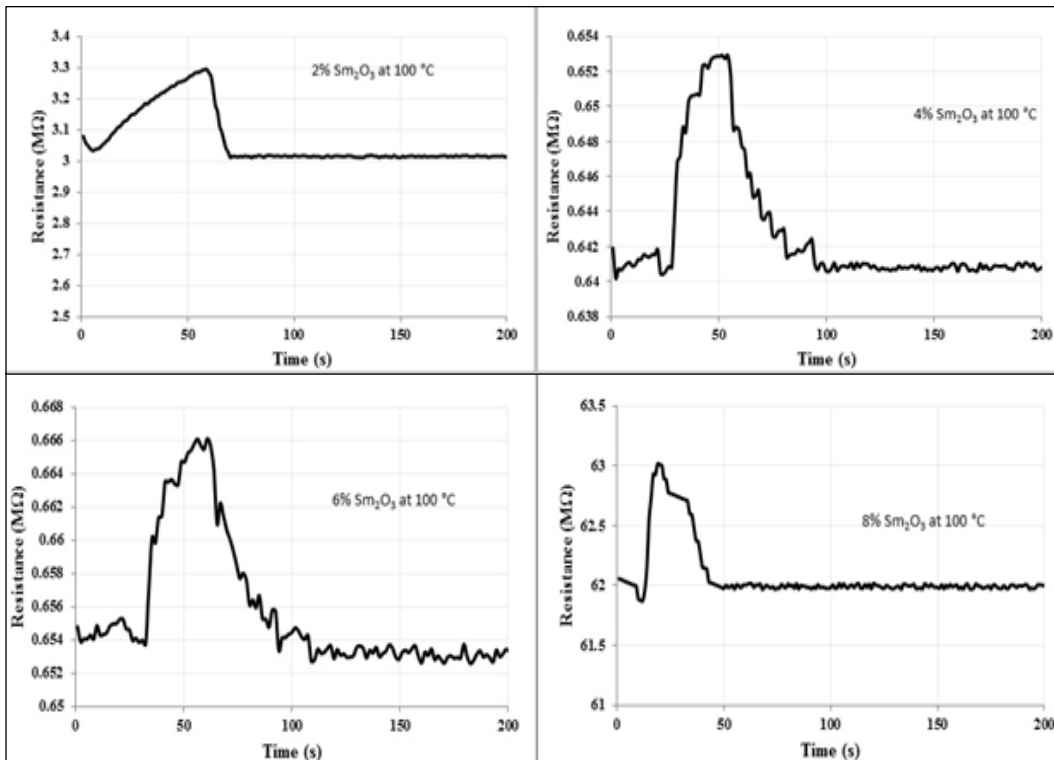


Figure 6. Variety for resistance as a function of time V₂O₅ films with (2, 4, 6, and 8) wt. % Sm₂O₃ ratio at 100 °C.



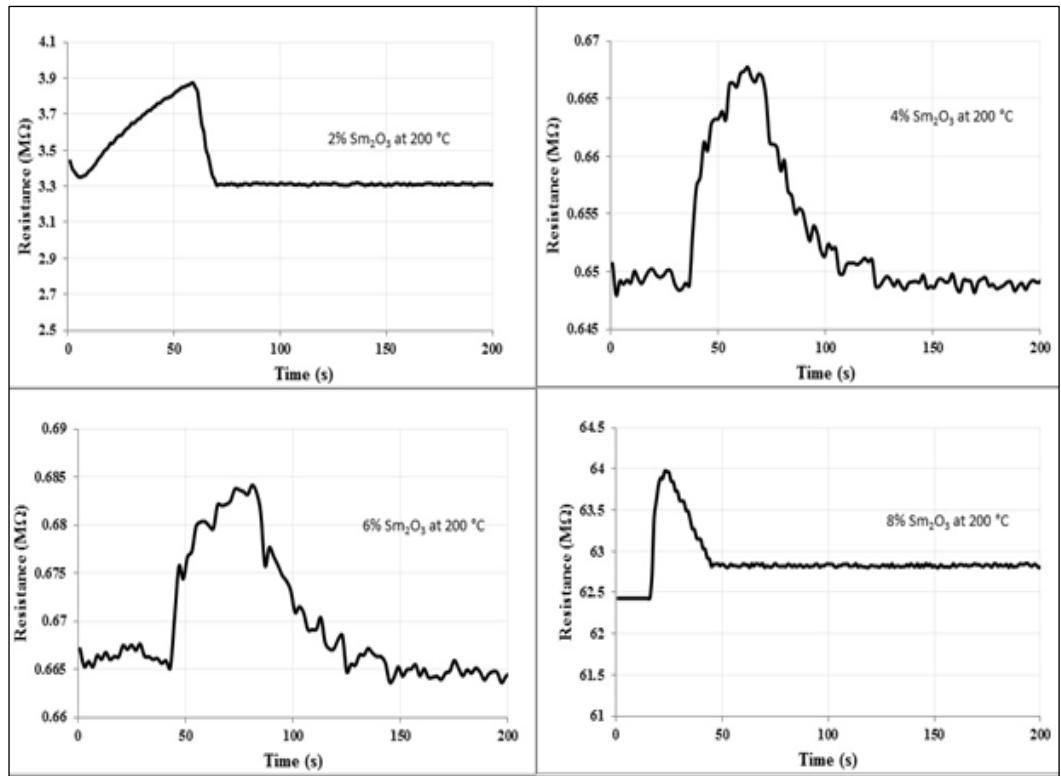


Figure 7. Variety for resistance as a function to the time V₂O₅ films with (2, 4, 6, and 8) wt. % Sm₂O₃ ratio at 200 C°
Table 2. Indicates the sensitivity, response and recovery time of V₂O₅ films with (2, 4, 6, and 8) wt. % Sm₂O₃ ratio

	Sensitivity%			Response			Recovery		
	50C°	100 C°	200 C°	50 C°	100 C°	200 C°	50 C°	100 C°	200 C°
PURE	96	96	95	18	8	21	27	7	23
2%	96	93	88	50	53	53	10	12	13
4%	99	98	97	10	14	25	18	19	43
6%	99	98	97	13	17	20	16	27	30
8%	99	98	98	8	10	8	30	21	21

Conclusions

V₂O₅:Sm₂O₃ thin films were prepared, using Pulsed Laser deposition method successfully. X-ray diffraction declared V₂O₅ monoclinic structure, as well as another phase V₄O₇ appeared at high doping ratio of Sm₂O₃, the appearance attributed to the abundance of oxygen supplied from the heat treatment as well as from the Sm₂O₃. FESEM images indicate homogeneous bunch distribution for columnar structure and the average grain size values decrease slightly for increment doping. The blue shift of E_g is attributed to the quantum confinement effect. The sensing behavior NO₂ gas of V₂O₅ doped with Sm₂O₃ /Si for NO₂ gas increases with doping ratio till reach steady value 99% give indication that the surface of gas sensors cells reach saturation, the response time equivalent to 10s at doping ratio 8% at working temperature 50 °C.

References

Benmoussa M, Outzourhit A, Bennouna A, Ameziane EL. Electrochromism in sputtered V₂O₅ thin films: structural and optical studies. *Thin Solid Films* 2002; 405(1-2): 11-16.
 Zou CW, Gao W. Fabrication, optoelectronic and photocatalytic properties of some composite oxide nanostructures. *Transactions on electrical and electronic materials* 2010; 11(1): 1-10.
 Londero E, Schröder E. Role of van der Waals bonding in the layered oxide V₂O₅: First-principles density-functional calculations. *Physical Review B* 2010; 82(5): 054116.
 Ali IM, Rzaij JM, Abbas QA, Ibrahim IM, Alatta HJ. Structural, optical and sensing behavior of neodymium-doped vanadium pentoxide thin films. *Iranian Journal of Science and Technology, Transactions A: Science*, 2018; 42(4): 2375-2386.
 Bhushan B. *Hand Book of Nanotechnology*. Springer 2007.
 Bochenkov VE, Sergeev GB. Sensitivity, selectivity, and stability of gas-sensitive metal-oxide nanostructures. *Metal oxide nanostructures and their applications* 2010; 3: 31-52.
 Gibbs ZM, LaLonde A, Snyder GJ. Optical band gap and the Burstein–Moss effect in iodine doped PbTe using diffuse



- reflectance infrared Fourier transform spectroscopy. *New Journal of Physics* 2013; 15(7): 075020.
- Zhao J, Ni J, Zhao X, Xiong Y. Preparation and characterization of transparent conductive zinc doped tin oxide thin films prepared by radio-frequency magnetron sputtering. *Journal of Wuhan University of Technology-Mater Science Education* 2011; 26(3): 388-392.
- Isam MI, Abubaker SM, Asmiet R. Energy Band Diagram of NiO: Lu₂O₃/n-Si heterojunction. *Iraqi Journal of Science* 2018; 59(1B): 287-293.
- Filipovic L, Selberherr S, Mutinati GC, Brunet E, Steinhauer S, Köck A, Grogger W. A method for simulating spray pyrolysis deposition in the level set framework. *Engineering Letters* 2013; 21(4): 224-240.
- Marchionni M, Caramel S, Stagnaro S. Inherited real risk of schizophrenia: Pathogenesis, bedside diagnosis and primary prevention. *NeuroQuantology* 2019; 17(5): 10-16.
- Saniotis A. Becoming animals: Neurobiology of shamanic shapeshifting. *NeuroQuantology* 2019; 17(5): 81-86.

